

Metamorphic InAs 2DEGs for quantum computation platforms

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The development of quantum computing still relies on the development of advanced material platforms. Among the most promising candidates are semiconductor-superconductor hybrid systems, such as Andreev quantum bits and Kitaev transmons. These systems are based on high-quality superconducting thin films with transparent interfaces to low-dimensional semiconductors, offering the potential for extended coherence times and robust qubit-qubit coupling [1].

To this end, a metamorphic growth protocol has been employed, and low-temperature electron mobilities up to $8 \times 10^5 \text{ cm}^2/\text{Vs}$ have been achieved in undoped deep InAs/In_{0.81}Ga_{0.19}As two-dimensional electron gases (2DEGs) grown on GaAs (001) [2]. Additionally, superconducting proximity effects have been observed in Josephson junctions between shallow InAs 2DEGs and epitaxial Al layers [3]. Optimal mobilities were achieved by tuning the thickness t of a strain-relieving In_{0.84}Al_{0.16}As layer beneath the quantum well (QW) region [2].

Here, we discuss the strain relaxation dynamics for varying t and their impact on electron scattering mechanisms in the InAs 2DEGs. Two-dimensional XRD reciprocal space maps of the (004) and (224) reflections (Fig. 1) reveal how increasing t from 50 nm to 300 nm leads to near-complete strain relaxation and a reduction in mosaicity in both the InAs QW and the surrounding barriers. Mobility measurements from gated Hall bars (Fig. 2) reveal striking gains in electron mobility and a reduced anisotropy between the [110] and [-110] orientations as t grows. This improvement stems from diminished anisotropic scattering mechanisms, linked to the cross-hatch roughness pattern—a memory of the buried dislocation network in the buffer layer (see insets of Fig. 2). These features, shaped by strain and composition fluctuations, highlight the interplay between structural engineering and electronic properties.

[1] J. S. Lee et al. *Nano Lett.* **19**, 3083 (2019)

[2] A. Benali et al., *J. Cryst. Growth* **593**, 1267681 (2022)

[3] M. Sütő et al. *Phys. Rev. B* **106**, 235404 (2022)

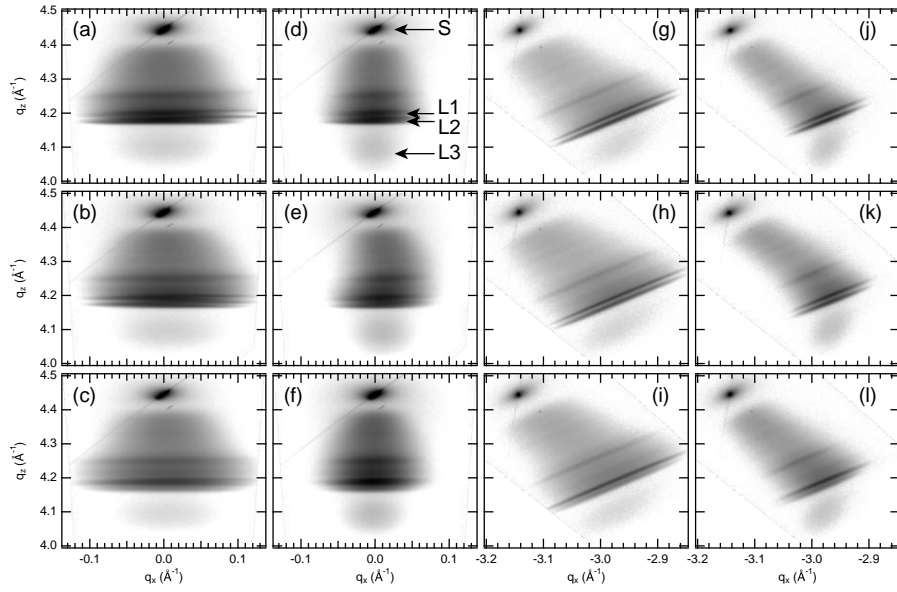


Fig. 11: X ray diffraction reciprocal space maps of (004) (a-f) and (224) reflections (g-l) of InAs QWs along the [110] direction (a-c, g-i) and along the [-110] direction (d-f, j-l) with $t = 300$ nm (a,d,g,j) , 150 nm, (b,e,h,k) and 50 nm (c,f,i,l).

Substrate, $\text{In}_{0.81}\text{Al}_{0.19}\text{As}$, $\text{In}_{0.84}\text{Al}_{0.16}\text{As}$, and InAs peaks are labelled as S, L1, L2 and L3, respectively.

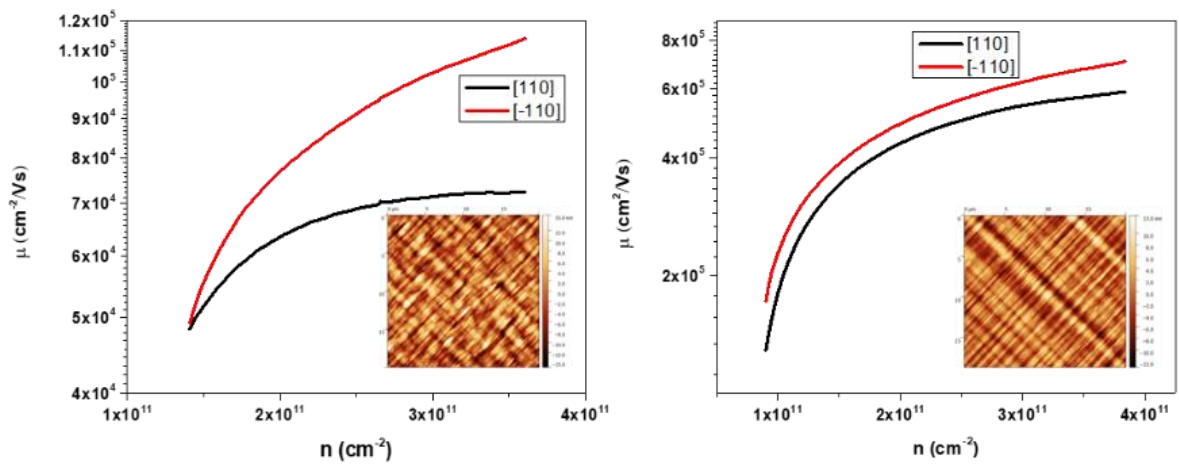


Fig. 22: Mobility vs. charge density along [110] and [-110] for $t = 50$ nm (left) and 300 nm (right). Inset: corresponding AFM images of top surfaces.

